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## Crystal Structure

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# The high-pressure form of cadmium vanadate, $\mathrm{CdV}_{2} \mathrm{O}_{6}$ 

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The crystal structure of a new high-pressure modification of cadmium divanadium hexaoxide, $\mathrm{CdV}_{2} \mathrm{O}_{6}$, was refined from X-ray single-crystal data. It contains zigzag chains of edgesharing $\mathrm{VO}_{6}$ octahedra. Octahedra in adjacent chains share corners and form corrugated layers. Octahedrally coordinated Cd atoms, which lie on twofold axes, are situated between the layers. The columbite-like structure results in a strong distortion of the $\mathrm{CdO}_{6}$ octahedra which may be stabilized only at high pressure.

## Comment

Complex vanadium $(\mathrm{V})$ oxides have received renewed interest as electrode materials and catalysts. A number of compounds with the general formula $M V_{2} \mathrm{O}_{6}(M=\mathrm{Mg}, \mathrm{Ca}, \mathrm{Mn}, \mathrm{Co}, \mathrm{Ni}, \mathrm{Cu}$, Zn and Cd ) have been studied in the last decade (Rao \& Palanna, 1996; Fuentes et al., 1999; Wei et al., 2005, 2006). Among these compounds, $\mathrm{CuV}_{2} \mathrm{O}_{6}$ (Cao et al., 2006) and $\mathrm{CdV}_{2} \mathrm{O}_{6}$ (Chen et al., 2004) have been shown to be promising cathode materials for rechargeable lithium batteries. Numerous $M \mathrm{~V}_{2} \mathrm{O}_{6}$ oxides exist at ambient pressure (Rao \& Palanna, 1996; Mocala \& Ziolkowski, 1987; Tsuzuki et al., 1989) and their crystal structures have been reported by different authors for $M=\mathrm{Mg}$ ( Ng \& Calvo, 1972), Ca (Bouloux et al., 1972), Mn (Mueller-Buschbaum \& Kobel, 1991a), Co (Jasper-Toennies \& Mueller-Buschbaum, 1984; Mueller-Buschbaum \& Kobel, 1991a), Ni (LeBail \& Lafontaine, 1990; Mueller-Buschbaum \& Kobel, 1991b), Cu (Lavaud \& Galy, 1972; Calvo \& Manolescu, 1973), Zn (Angenault \& Rimsky, 1968; Andreetti et al., 1984) and Cd (Bouloux \& Galy, 1969; Bouloux et al., 1972). Although the symmetry of the $M \mathrm{~V}_{2} \mathrm{O}_{6}$ structures at ambient pressure is changed from orthorhombic to triclinic, all of them belong to the brannerite or thorutite $\left(\mathrm{ThTi}_{2} \mathrm{O}_{6}\right)$ structure type. Except for $M=\mathrm{Mn}$, $M \mathrm{~V}_{2} \mathrm{O}_{6}$ transforms at high pressure to the orthorhombic columbite $\left(\mathrm{FeNb}_{2} \mathrm{O}_{6}\right)$ structure type (Gondrand et al., 1974). The high-temperature polymorph of $\mathrm{HgV}_{2} \mathrm{O}_{6}$ also has a
columbite-like structure (Mormann \& Jeitschko, 2000), which is the only columbite-like $M \mathrm{~V}_{2} \mathrm{O}_{6}$ vanadate stable at ambient pressure.

High-pressure polymorphs are usually denser (6-10\% for $M \mathrm{~V}_{2} \mathrm{O}_{6}$ phases) than their ambient pressure counterparts, and often have layered or close-packed structures. Layered structures are quite promising for applications as cathode materials. Therefore, the investigation of the high-pressure $M \mathrm{~V}_{2} \mathrm{O}_{6}$ polymorphs in addition to the ambient pressure modifications may also be useful for their possible applications. However, the crystal structures of the $M V_{2} \mathrm{O}_{6}$ compounds at high pressure have not been investigated yet, although their X-ray patterns have been reported (Gondrand et al., 1974). In this paper, we report the preparation of single crystals of a high-pressure modification of $\mathrm{CdV}_{2} \mathrm{O}_{6}$ and its crystal structure.

Two types of $\mathrm{MO}_{6}$ octahedra are present in the title $\mathrm{CdV}_{2} \mathrm{O}_{6}$ structure. The $\mathrm{VO}_{6}$ octahedra are connected via edge sharing and form zigzag chains running along the $a$ direction (Fig. 1a). Octahedra in adjacent chains share corners and form corrugated layers separated by layers of $\mathrm{Cd}^{2+}$ cations; the Cd atoms lie on twofold axes. The $\mathrm{VO}_{6}$ octahedron is strongly distorted (Fig. 2b) and the V atom is shifted towards atom O3 away from atoms O1 and O2, due to the formation of a vanadyl bond [1.6548 (10) $\AA$ ]. As a result, the $\mathrm{V}-\mathrm{O} 2$ bond opposite the vanadyl bond is elongated to 2.3505 (10) $\AA$. The $\mathrm{O} 2-\mathrm{V}-\mathrm{O} 3$ angle is 167.72 (4) ${ }^{\circ}$. Bond-valence-sum (BVS) calculations give values of 5.23 and 4.87 for six- and five-coordinated vanadium, respectively (Schindler et al., 2000). This type of distortion of the coordination polyhedron is typical for pentavalent vanadium, and similar distances and angles for the $\mathrm{VO}_{6}$ octahedron were found, for instance, in the columbitelike structure of $\mathrm{HgV}_{2} \mathrm{O}_{6}$ (Mormann \& Jeitschko, 2000). The $\mathrm{CdO}_{6}$ octahedra are linked by edge sharing into a zigzag chain running along the $a$ direction. The chains are not connected to each other (Fig. 1b). The $\mathrm{CdO}_{6}$ octahedra are also strongly distorted (Fig. 2a) and have three pairs of $\mathrm{Cd}-\mathrm{O}$ distances, viz. 2.1961 (10), 2.3071 (10) and 2.4311 (10) $\AA$. The largest $\mathrm{O}-\mathrm{Cd}-\mathrm{O}$ angle is $155.82(4)^{\circ}$. BVS calculation (Brown \& Altermatt, 1985; Brese \& O'Keeffe, 1991) gives a value of 2.06, in spite of the strong distortion of the polyhedron. One may


Figure 1
The crystal structure of $\mathrm{CdV}_{2} \mathrm{O}_{6}$, showing (a) corrugated layers of $\mathrm{VO}_{6}$ octahedra and $(b)$ chains of $\mathrm{CdO}_{6}$ octahedra.

(a)

(b)

Figure 2
Coordination polyhedra for $(a)$ the Cd atoms and $(b)$ the V atoms. [Symmetry codes as in Table 1; plus (v) $-x+\frac{3}{2}, y,-z+2$; (vi) $-x+\frac{3}{2}$, $y+1,-z+2$.]
speculate that this distortion is a consequence of the almost pyramidal coordination of the V atom.

One may suggest that the strong distortion of the coordination polyhedra in the high-pressure structure is a reason for the instability of this modification at ambient pressure. In the brannerite-like structure, the Cd atom has an almost regular octahedral coordination with a small difference between the equatorial and apical $\mathrm{Cd}-\mathrm{O}$ distances ( 2.287 and $2.199 \AA$, respectively). The coordination polyhedron of the V atom should be considered as a trigonal bipyramid with two vanadyl bonds of 1.687 and $1.708 \AA$ (Zavalij \& Whittingham, 1999). In the high-pressure structure, the V atoms have six O -atom neighbours. However, the sixth $\mathrm{V}-\mathrm{O} 2$ distance is $2.3505(10) \AA$, and it is possible to use the $\tau$ parameter (Addison et al., 1984) to distinguish between the $\mathrm{VO}_{5}$ squarepyramidal and trigonal-bipyramidal geometries. The $\tau$ parameter is 0.12 , indicating that, if the sixth O 2 atom is ignored, the $\mathrm{VO}_{5}$ polyhedron is a square pyramid. Thus, the coordination arrangement of the V atom in both $\mathrm{CdV}_{2} \mathrm{O}_{6}$ and $\mathrm{HgV}_{2} \mathrm{O}_{6}$ is a distorted octahedron. In $\mathrm{HgV}_{2} \mathrm{O}_{6}$, the $\mathrm{VO}_{6}$ polyhedron $(\tau=0.17)$ is very similar to that in $\mathrm{CdV}_{2} \mathrm{O}_{6}$ but the sixth V-O bond is longer ( $2.442 \AA$ ). The main difference between the two structures arises from the coordination of the Hg atom, with two very short $(2.040 \AA)$ and four long (2.643 and $2.677 \AA) \mathrm{Hg}-\mathrm{O}$ distances. Such a coordination stabilizes a columbite-like structure, in spite of the nearly square-pyramidal coordination of the V atom. One should note that in other known columbite-like compounds both cations have a nearly regular octahedral coordination. This is not the case for $\mathrm{CdV}_{2} \mathrm{O}_{6}$ and, although the cation-cation distances are very close in both modifications, the O -atom environments for these cations are significantly different. Thus, one may conclude that the $M \mathrm{~V}_{2} \mathrm{O}_{6}$ columbite-like structure results in a strong distortion of the $M \mathrm{O}_{6}$ octahedra which may be stabilized only at high pressure.

## Experimental

A mixture of CdO (99.9\%) and $\mathrm{V}_{2} \mathrm{O}_{5}(99.9 \%)$ in a $1: 1$ ratio was placed in a gold capsule and treated at 6 GPa and 1473 K in a belttype high-pressure apparatus for 60 min (heating rate of
$120 \mathrm{~K} \mathrm{~min}^{-1}$ ). After heat treatment, the sample was quenched to room temperature and the pressure was released slowly. The product consisted of small black single crystals suitable for X-ray crystal structure analysis. Powder X-ray diffraction revealed the crystals to be $\mathrm{CdV}_{2} \mathrm{O}_{6}$.

Crystal data
$\mathrm{CdV}_{2} \mathrm{O}_{6}$
$M_{r}=310.29$
Orthorhombic, Pnab
$a=4.9495$ (6) $\AA$
$b=5.6949$ (7) $\AA$
$c=14.1692$ (17) $\AA$

## Data collection

Enraf-Nonius CAD-4
diffractometer
Absorption correction: Gaussian
(JANA2000; Petříček \&
Dušek, 2000)
$T_{\text {min }}=0.314, T_{\text {max }}=0.565$
4388 measured reflections
Refinement
$R\left[F^{2}>2 \sigma\left(F^{2}\right)\right]=0.016$
$w R\left(F^{2}\right)=0.028$
$S=1.19$
1349 reflections

$$
\begin{aligned}
& V=399.39(8) \AA^{3} \\
& Z=4 \\
& \text { Mo } K \alpha \text { radiation } \\
& \mu=9.79 \mathrm{~mm}^{-1} \\
& T=293 \mathrm{~K} \\
& 0.22 \times 0.11 \times 0.06 \mathrm{~mm}
\end{aligned}
$$

1727 independent reflections 1349 reflections with $I>3 \sigma(I)$
$R_{\text {int }}=0.020$
2 standard reflections
frequency: 120 min
intensity decay: none

> 43 parameters
> $\Delta \rho_{\max }=0.58 \mathrm{e} \AA^{-3}$
> $\Delta \rho_{\min }=-0.49 \mathrm{e}^{-3}$

Table 1
Selected bond lengths ( $\AA$ ).

| $\mathrm{Cd}-\mathrm{O} 1$ | 2.1961 (10) | $\mathrm{V}-\mathrm{O} 2$ | 2.0225 (10) |
| :---: | :---: | :---: | :---: |
| $\mathrm{Cd}-\mathrm{O3}^{\text {i }}$ | 2.4311 (10) | $\mathrm{V}-\mathrm{O} 2^{\text {iv }}$ | 1.7494 (10) |
| $\mathrm{Cd}-\mathrm{OB}^{\text {ii }}$ | 2.3071 (10) | $\mathrm{V}-\mathrm{O} 2^{\text {iii }}$ | 2.3505 (10) |
| $\mathrm{V}-\mathrm{O} 1$ | 2.0472 (10) | $\mathrm{V}-\mathrm{O} 3$ | 1.6548 (10) |
| $\mathrm{V}-\mathrm{O} 1^{\text {iii }}$ | 1.7853 (10) |  |  |

Symmetry codes: (i) $x, y+1, z$; (ii) $-x+1,-y+1,-z+2$; (iii) $x+\frac{1}{2},-y+1, z$; (iv)
$x+\frac{1}{2},-y+\frac{1}{2},-z+\frac{3}{2}$.

The nonstandard setting of the Pbcn space group was used for better representation of the $\mathrm{CdV}_{2} \mathrm{O}_{6}$ structure, which is built of zigzag chains of $\mathrm{VO}_{6}$ octahedra arranged in crimped layers. Thus, the entire structure may be considered as stacking of $\mathrm{V}_{2} \mathrm{O}_{6}$ and Cd layers along the $c$ axis. This was the main reason for choosing the nonstandard setting of the space group. For example, $\mathrm{Bi}_{2+x} \mathrm{Sr}_{2-x} \mathrm{CuO}_{6+d}$ crystallizes in the $A 2 / a$ space group, with cell parameters about $a=5.4 \AA, b=$ $5.4 \AA$ and $c=24.6 \AA$, and $\alpha=\beta=\delta=90^{\circ}$, which corresponds to the standard setting $C 2 / c$. Nevertheless, it is always refined in the $A 2 / a$ space group for better presentation of layer stacking.

The structure was refined isotropically to $R=0.032$ and anisotropically to $R=0.016$. Difference Fourier syntheses revealed the highest residual peaks in the vicinity of the Cd and V sites. Anharmonic displacement parameters based on the Gram-Charlier expansion of the structure factor were refined up to the fourth order. Although the depths of the negative regions in the probability density function maps were about $3-6 \%$, only a few parameters had a significance level above $3 \sigma$ and the reduction of the $R$ factor was about 0.001 . Therefore, in the final refinement cycles, all atoms were refined anisotropically. At the final stage, the occupancies of all sites were refined in turn, with deviations from unity less than $3 \sigma$ in all cases. Fixed full occupancies were then used for all atoms. The use of the $I>2 \sigma(I)$ cut-off instead of $I>3 \sigma(I)$ added only eight extra observed reflections and raised the $R$ index only slightly (0.018).

Data collection: CAD-4 Manual (Enraf-Nonius, 1988); cell refinement: CAD-4 Manual; data reduction: JANA2000 (Petríček \&

## inorganic compounds

Dušek, 2000); program(s) used to solve structure: SIR2002 (Burla et al., 2000); program(s) used to refine structure: JANA2000; molecular graphics: ATOMS for Windows (Dowty, 1998); software used to prepare material for publication: JANA2000.

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Supplementary data for this paper are available from the IUCr electronic archives (Reference: BC3043). Services for accessing these data are described at the back of the journal.

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